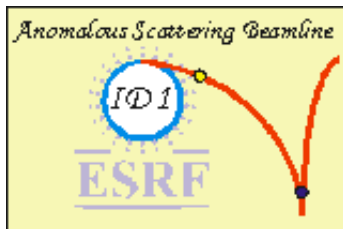


Coherent Diffraction Imaging as a tool to investigate the mechanics of Polycrystals

MID workshop, 2009 Grenoble

N. Vaxelaire , S. Labat , O. Thomas

IM2NP, Marseille



A. Diaz, T. H. Metzger



V. Jacques,
F. Picca, S. Ravy



H. Proudhon, S. Forest



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C. Kirchlechner



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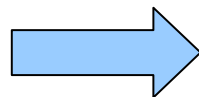
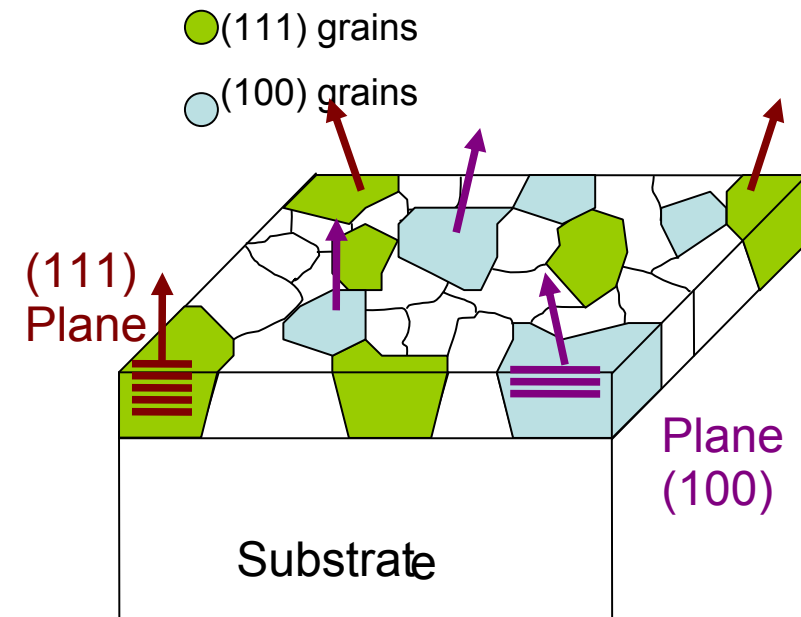


CENTRE NATIONAL
DE LA RECHERCHE
SCIENTIFIQUE



Context and Purpose

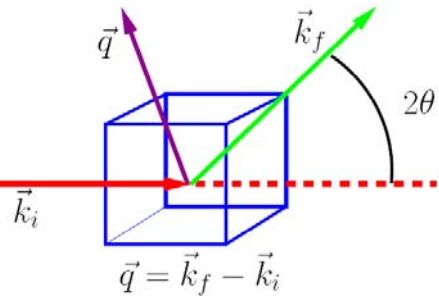
- The mechanical properties of small objects play a key role in numerous phenomena
- The local behaviour of Polycrystals is being still debated
- Strong lack of experimental techniques to probe deformation on this nanometer scale
- Coherent diffraction : a promising tool



Single grain behaviour

Phase Retrieval: specific case of strain objects

Kinematic theory of diffraction



$$I(\mathbf{q}) = \left| \sum_n f_n(\mathbf{q}) \exp i\mathbf{q}\mathbf{r}_n \right|$$

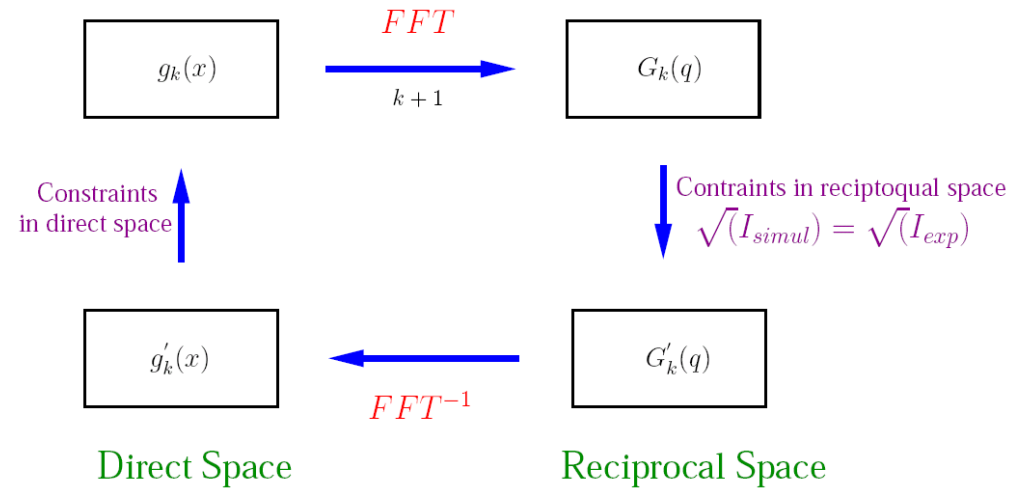
Diffracted Intensity

Displacement field

$$\mathbf{r}_n = \mathbf{R} + \mathbf{u}$$

$$I(\mathbf{q}) \propto |TF \{ \rho(\mathbf{r}) \cdot \exp(i\mathbf{G} \cdot \mathbf{u}(\mathbf{r})) \}|^2$$

Iterative Phase Retrieval Algorithm



Difficulties

- Tight support is needed for good reconstruction
- Initial support cannot be determined from autocorrelation (Patterson)
- Convergence more difficult (add condition of density)

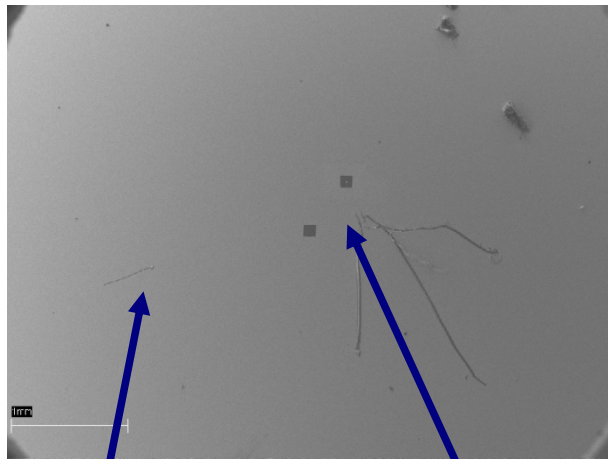
Our model sample

Main features

- Gold sample
- Special design

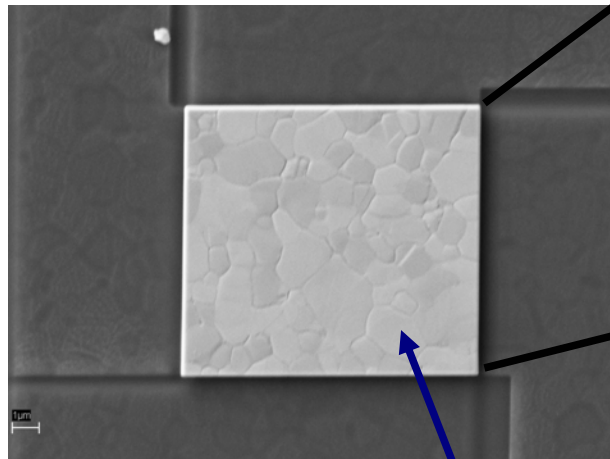
Why

- Good diffraction power, no corrosion issue during heating
- Find which single grain is diffracting



Homogeneous film
(375nm thick)

150µm square
area drilled by FIB



10*10µm
polycrystal



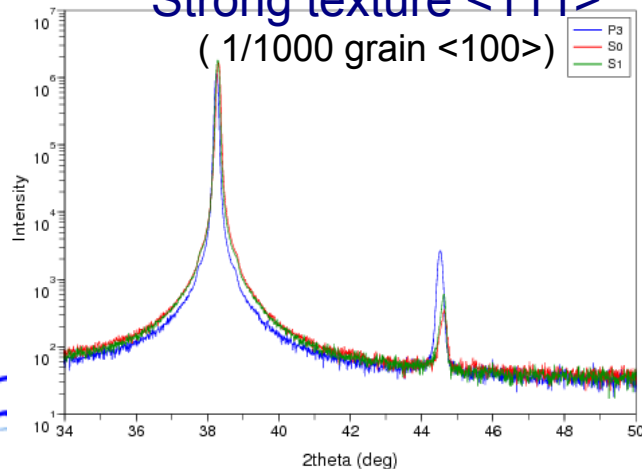
133 grains



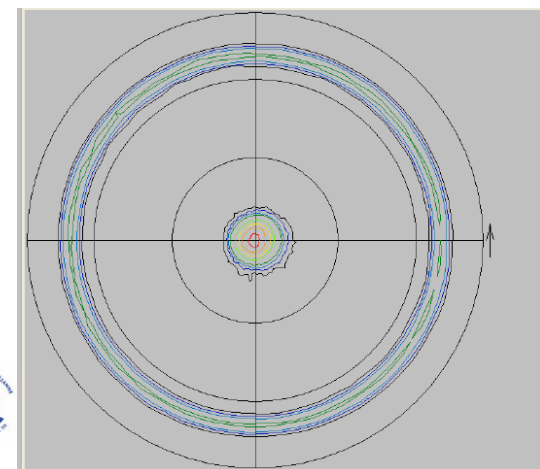
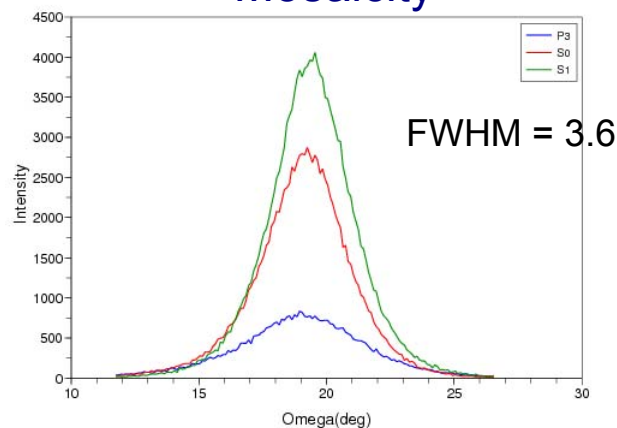
Single grain
behaviour

In-plane random
orientation

Strong texture $\langle 111 \rangle$
(1/1000 grain $\langle 100 \rangle$)

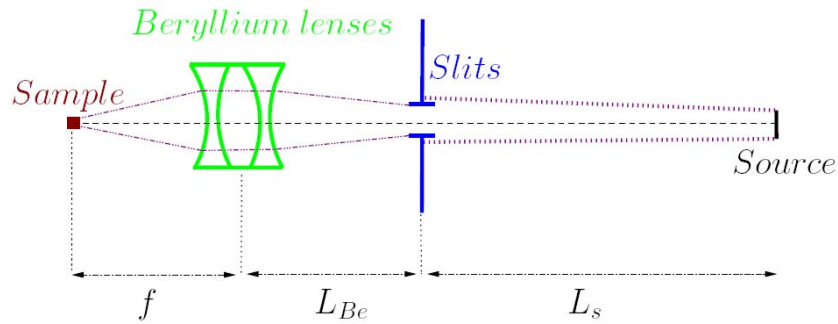


Mosaicity



Set up used on ID01 Beamline

Beam characteristics



Focal length ~ 40cm
 Beam size ~ $1 \times 3 \mu\text{m}$
 Divergence ~ $50 \mu\text{rad}$

Focused beam



Monochromatic beam

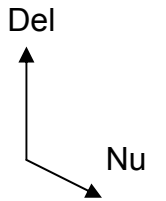
$E = 8,10 \text{ keV}$
 $\lambda = 1,53 \text{ \AA}$

Undulator source

(111) Bragg peak

$$2\theta_{111} = 37,8^\circ$$

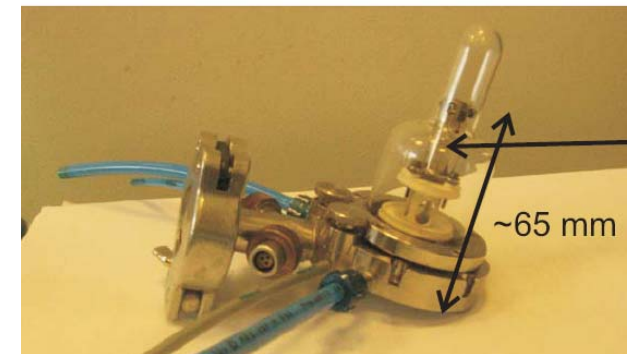
Projected Beam footprint on sample:
 $3 \times 3 \mu\text{m}$



CCD camera

- Indirect illuminated deep depletion detector
- Pixel size : $22 \mu\text{m}$

Furnace



Set up used on ID01 Beamline

Beam

Sample

Focal
Beam
Diver

Nu

mera
ed deep

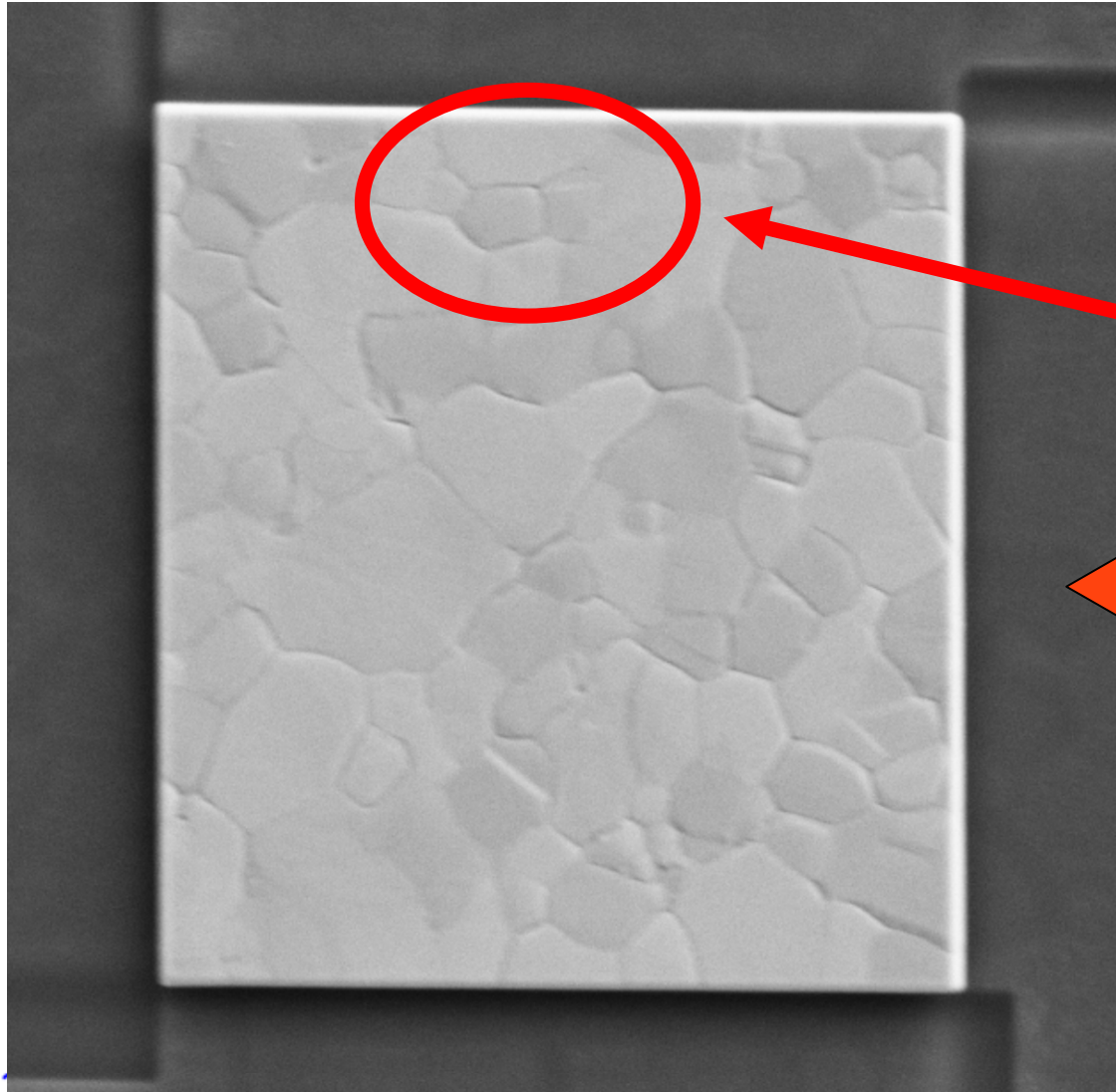
Undula
source

65 mm

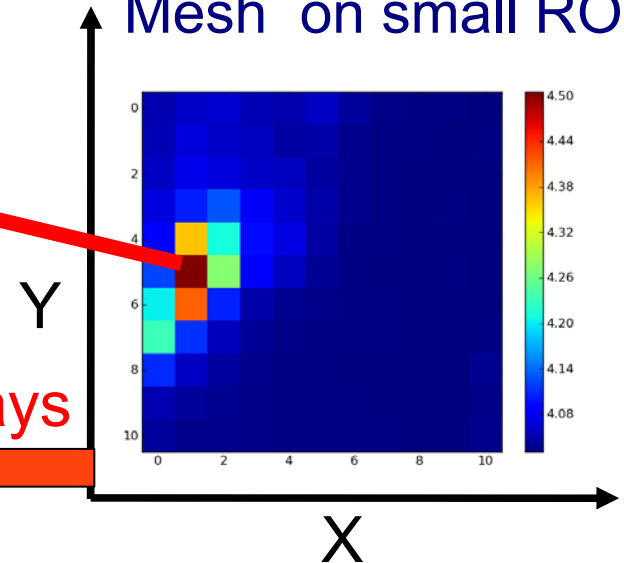
Grain determination

Scanning 3DXRD approach $\{x,y\} * \{\text{Nu}, \text{Del}, \text{Omega}\}$

Granular Debye ring Angular Map – Pole



Direct space
Mesh on small ROI



X-rays

Projected grain

seen

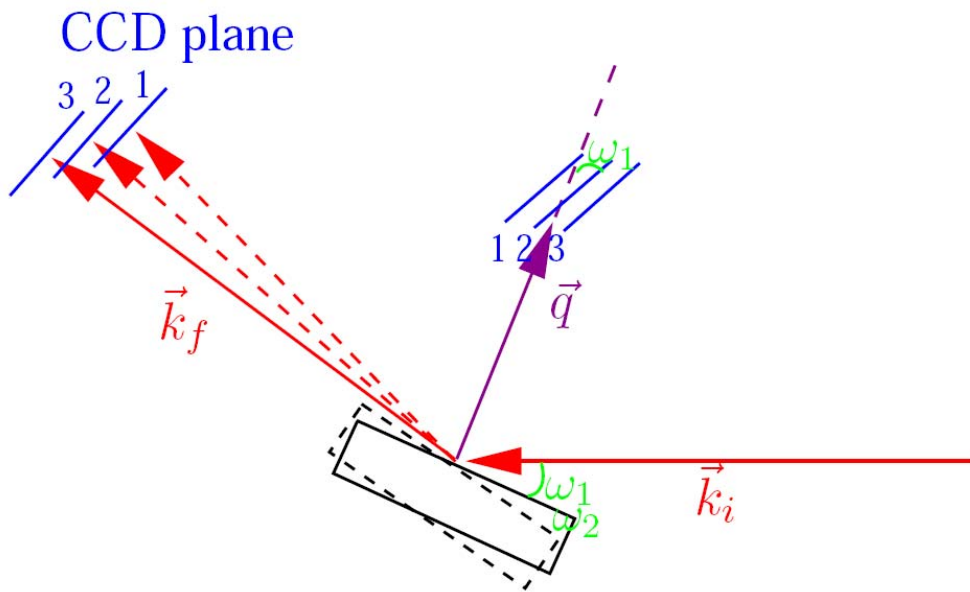
s

Projected
Beam size :
3*3 μm

3D Bragg Peak acquisition

Rocking curve peak (111)

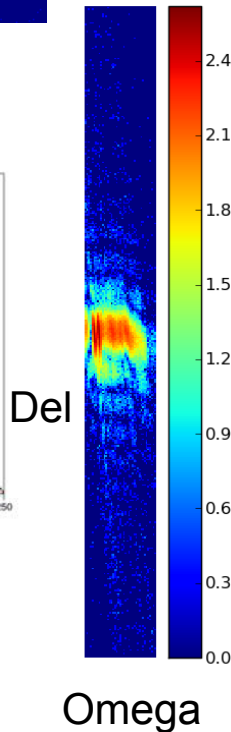
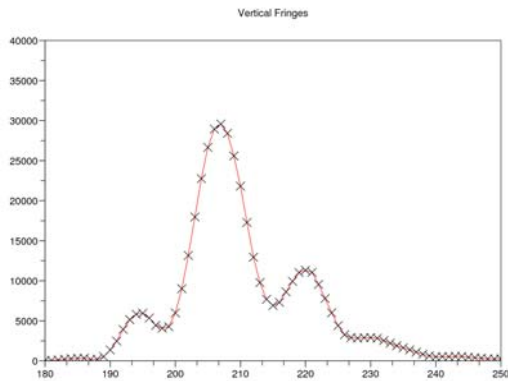
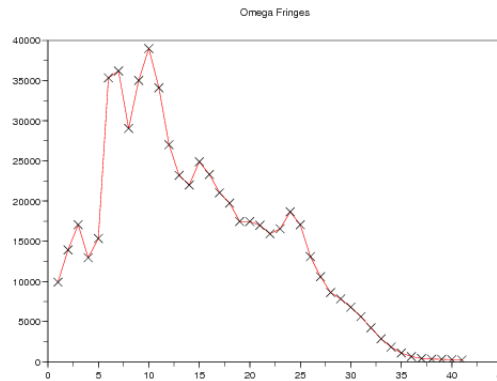
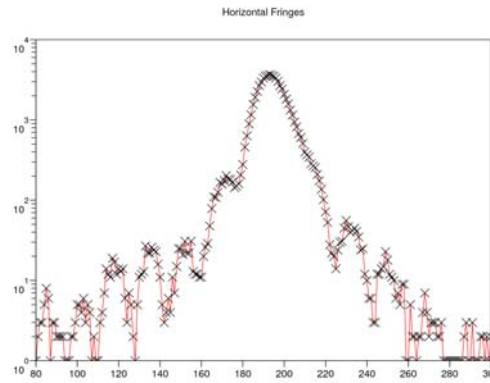
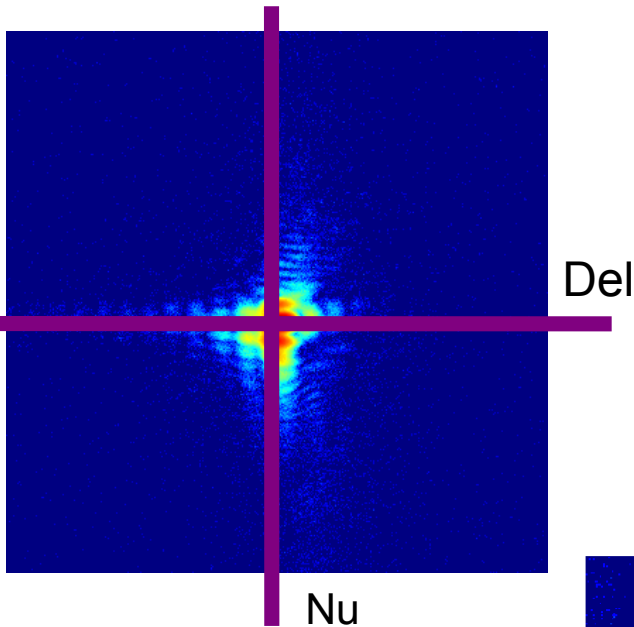
- Diffraction geometry
- Oblique cut in reciprocal space
- Weak resolution in Omega direction



Acquisition characteristics

- Detector Indirect Deep depletion CCD 22 μ m pixels
- Single photon counting mode
- 3D acquisition time = 8 hours
- Max photons = 11 957 on one pixel
- Sum photon (all Bragg Peaks) = 32 .10⁶ photons

Oversampling and Characteristic dimensions

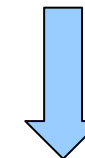


Reciprocal Space Resolution

- Δq (ccd) = $1,7 \text{ e-}4 \text{ nm-}1$
- Δq (omega) = $1,118\text{e-}3 \text{ nm-}1$

3D Oversampling

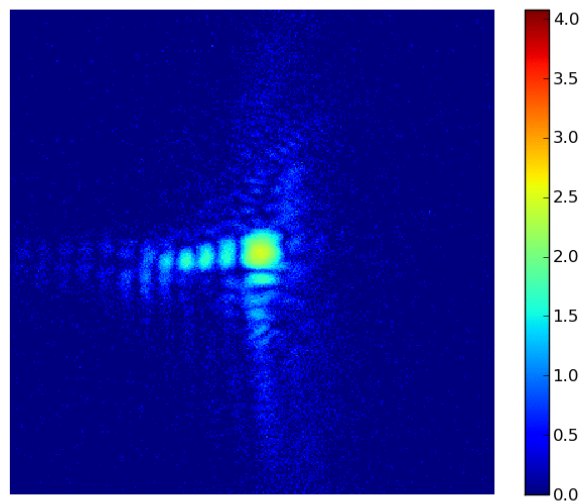
- $S_x = 15$
- $S_y = 8$
- $S_z = 4$



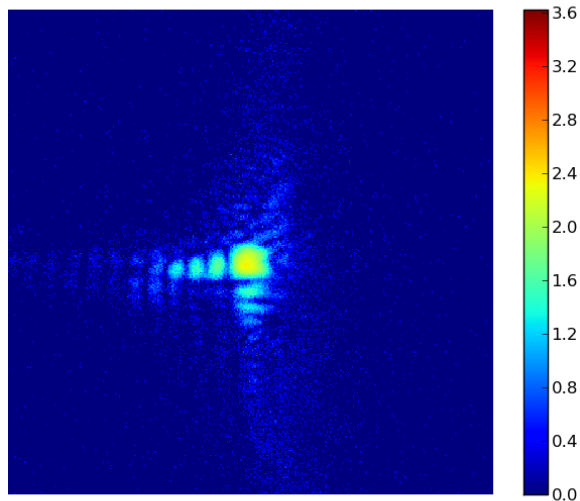
Dimension and Resolution in direct space

- $X = 738 \text{ nm}$ [$\Delta(X) = 15.4 \text{ nm}$]
- $Y = 230 \text{ nm}$ [$\Delta(Y) = 20 \text{ nm}$]
(Beam direction)
- $Z = 394 \text{ nm}$ [$\Delta(Z) = 15.4 \text{ nm}$]
(Film thickness)

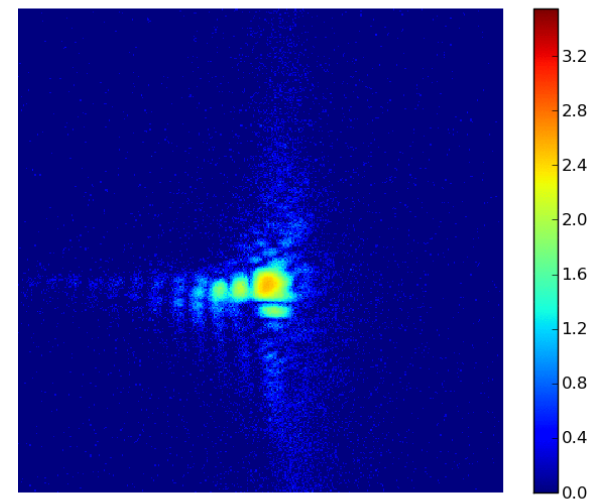
T=50deg



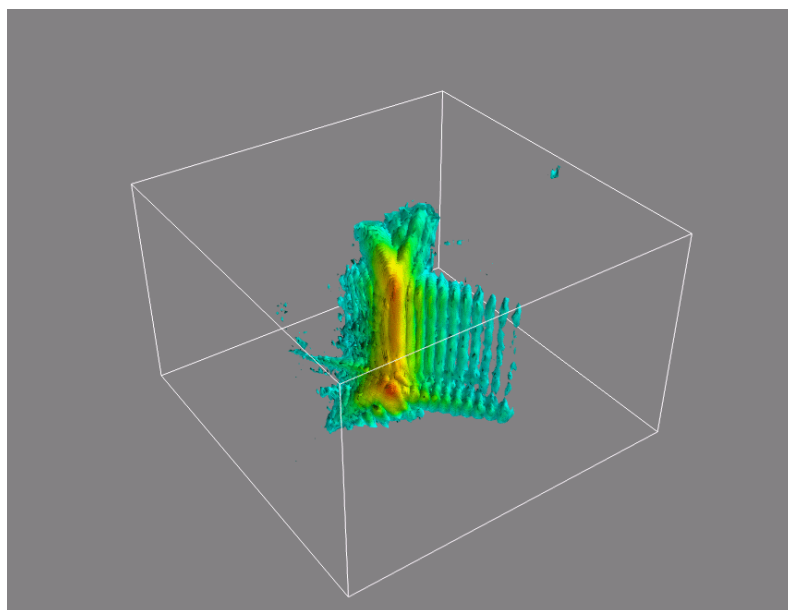
T=100deg



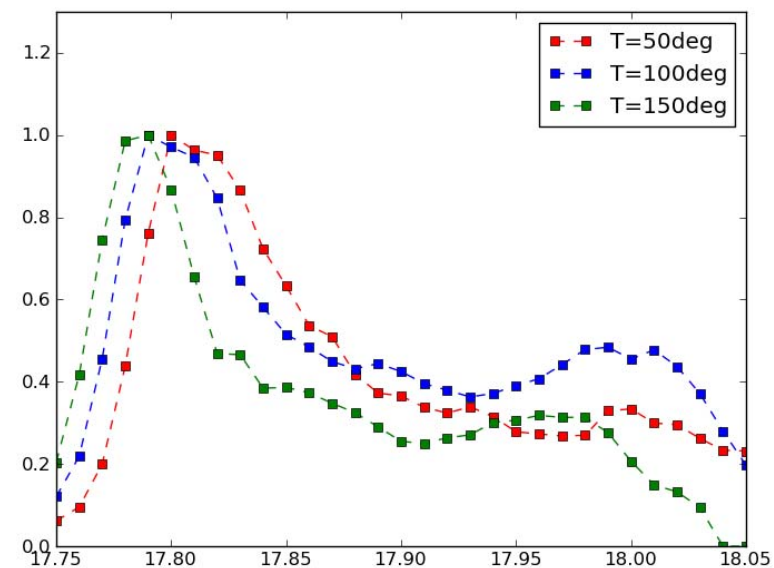
T=150deg



T=100deg

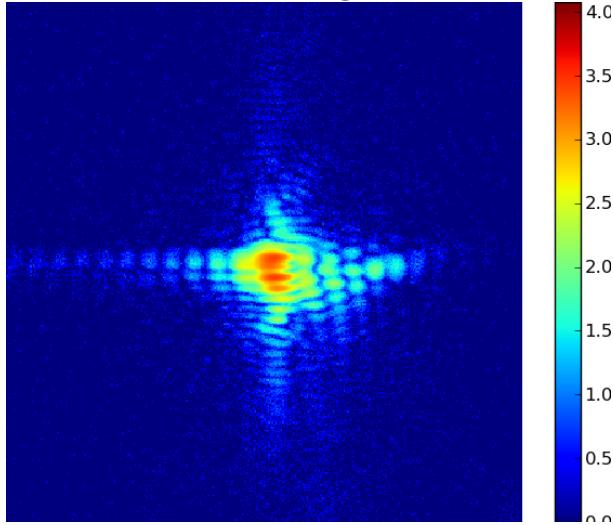


Rocking curve comparison

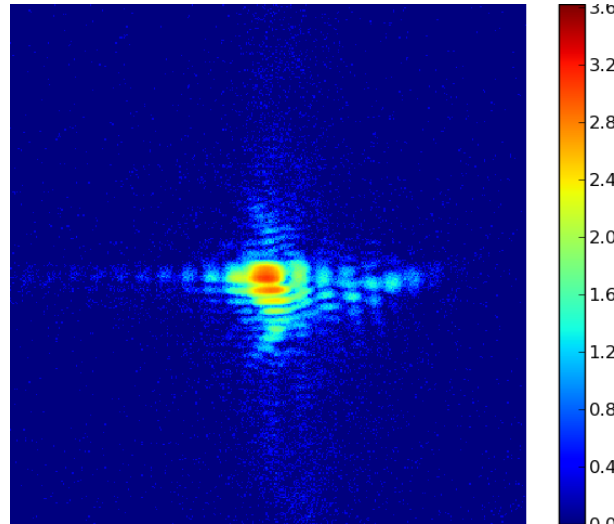


Experimental results

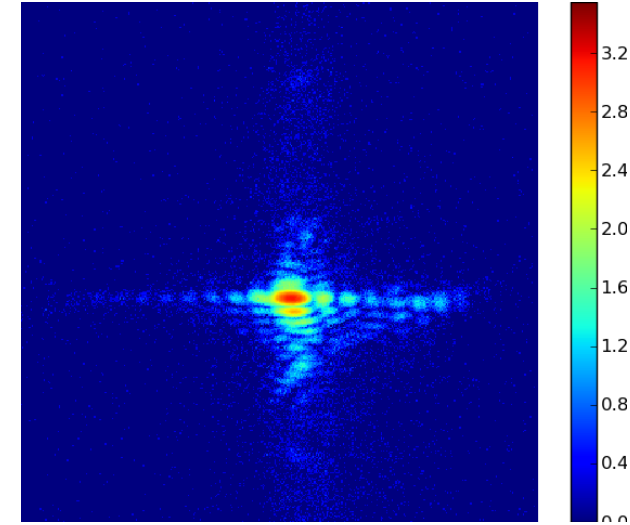
T=50deg



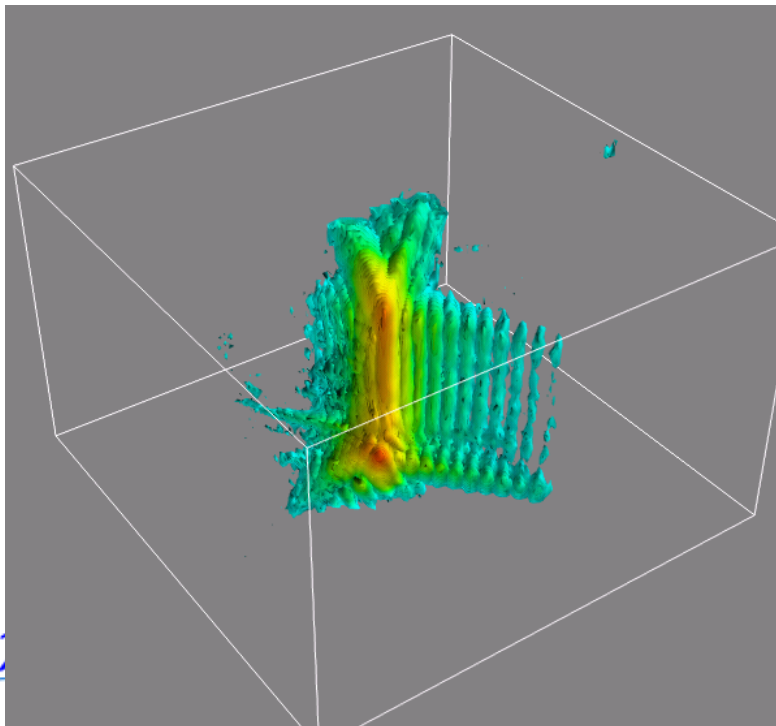
T=100deg



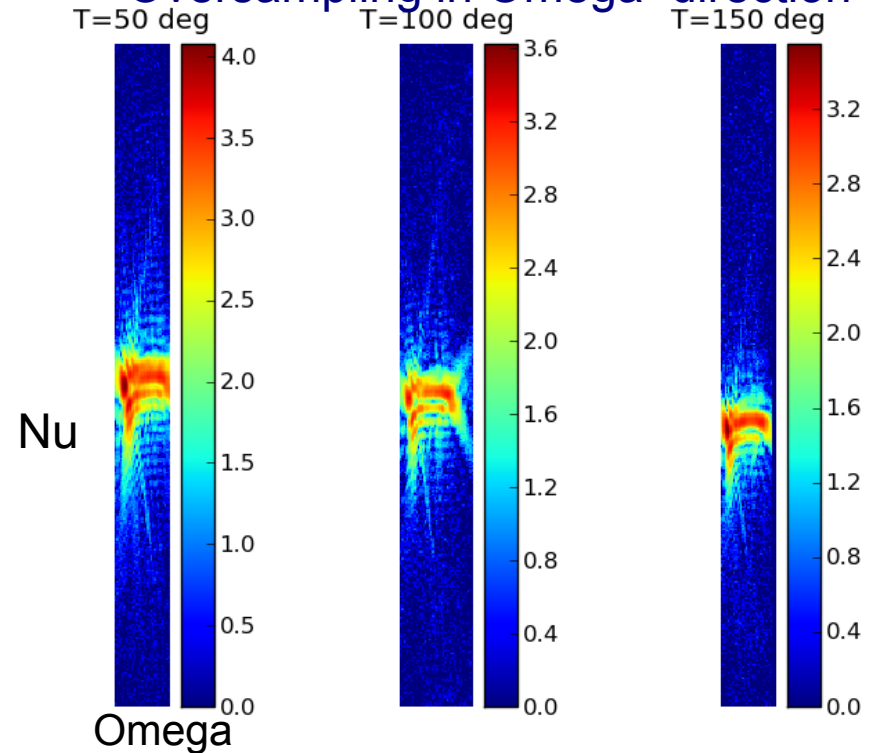
T=150deg



3D representation at T=100Deg



Oversampling in Omega direction



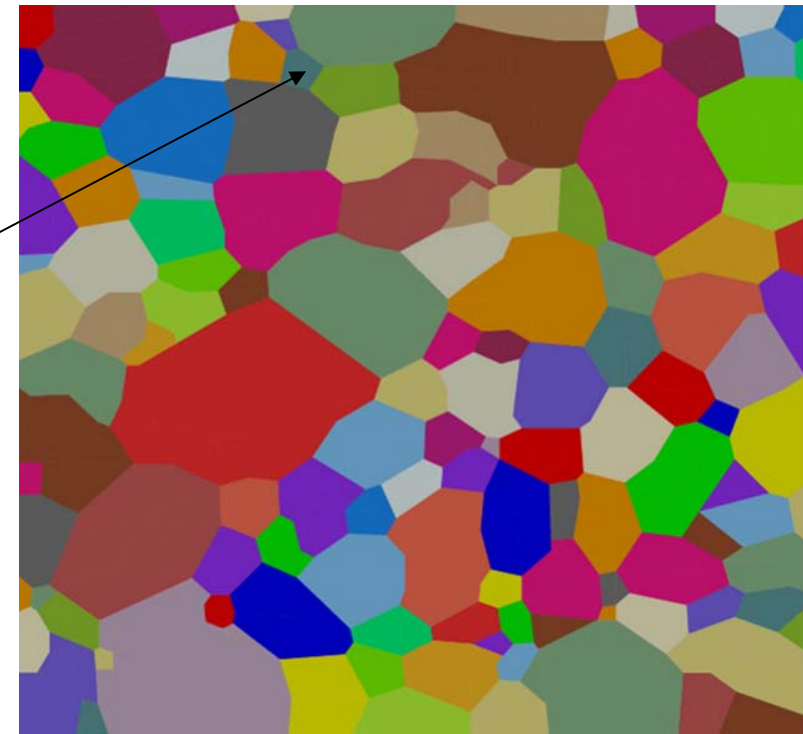
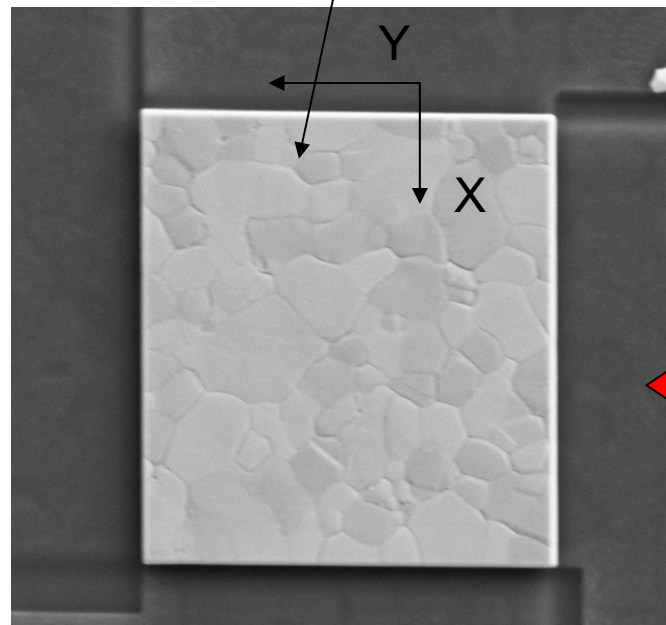
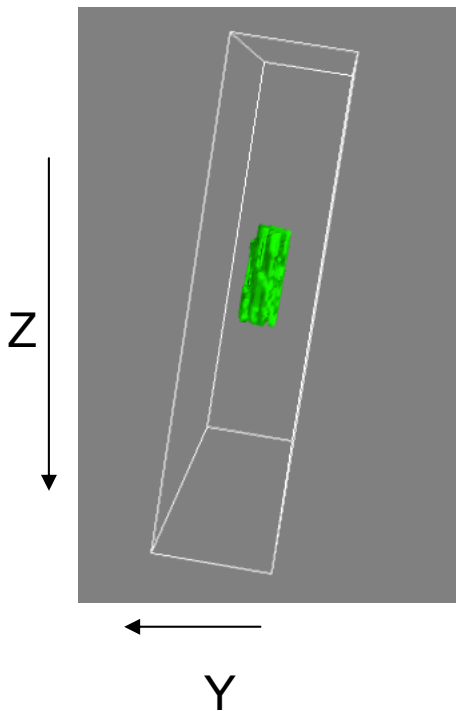
First reconstructions

Retrieval Procedure

- Initial Support=cube estimated from fringe spacing
- 5 shrinkwraps
- Each Cycle 5*{750ER+250HIO }
- = 50 000 FFT (50 hours)

Shape

Reconstructed shape



X-rays



Grain raw dimensions

X= 750nm

Y= 240 nm

Z= 390 nm

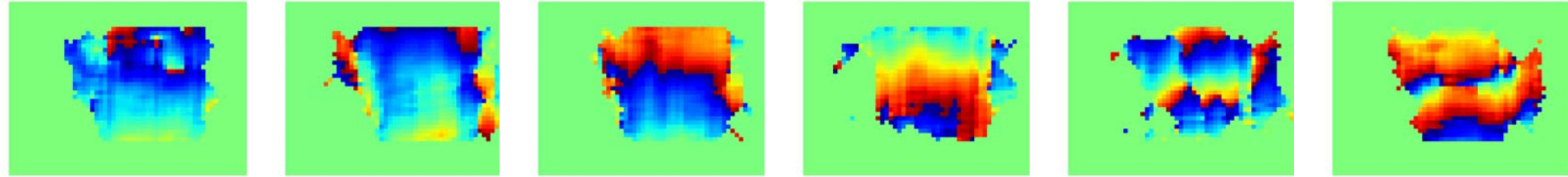
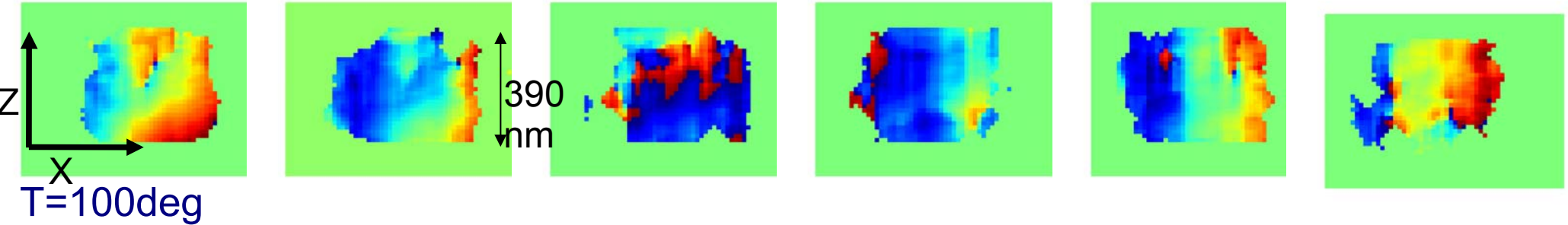
3D Phase in direct space : $G.u$ (Y perpendicular plane)

T=50deg

$G.u = G_{uz}(x,y,z)$

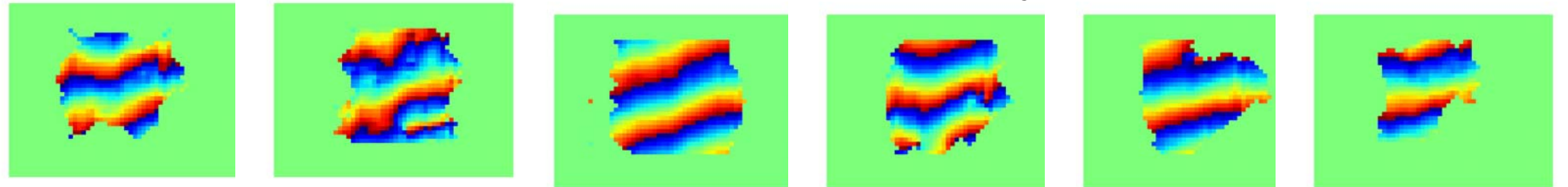
730 nm

390 nm



T=150deg

3 phase jump



Progressive apparition of a gradient in Z direction



complementary calculations under progress

Xfel Perspectives

SASE1 Beamline

Parameter	Unit	Value
Photon energy	keV	12.4
Pulse duration	fs	<100
Bandwidth	0.1 %	0.8
Coherence time	fs	0.2
Divergence	μrad	1
Source size	μm	70
Source distance	m	1000
Photons per pulse	#	10^{12}
Pulse brilliance	B	5×10^{33}

Coherent longitudinal length = 60 nm

could be improved
(Monochromator)

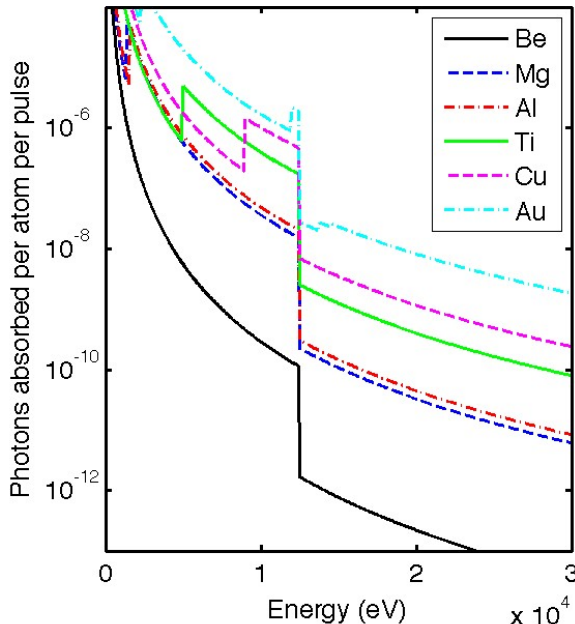
Comparison

Photons/s = 10^{12} ID01
focused beam

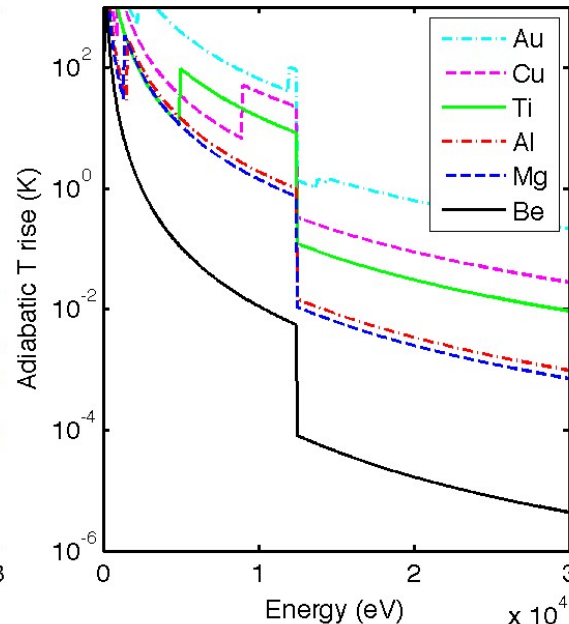
Au sample

- At 12.4 KeV 10% absorption for 400nm thick sample
- 1 pulse : 20 000 deg C

(a) 0.00014 BW, 1000000 μm^2 focus



(b) 0.00014 BW, 1000000 μm^2 focus



Aluminum polycrystal

Evaluation of thermal relaxation
between two pulses

New possibilities ?

- Dynamic of defect (dislocations)
- Grain growth
-

Thank You For Your Attention

First reconstructions

3D Phase in direct space : $G \cdot u$

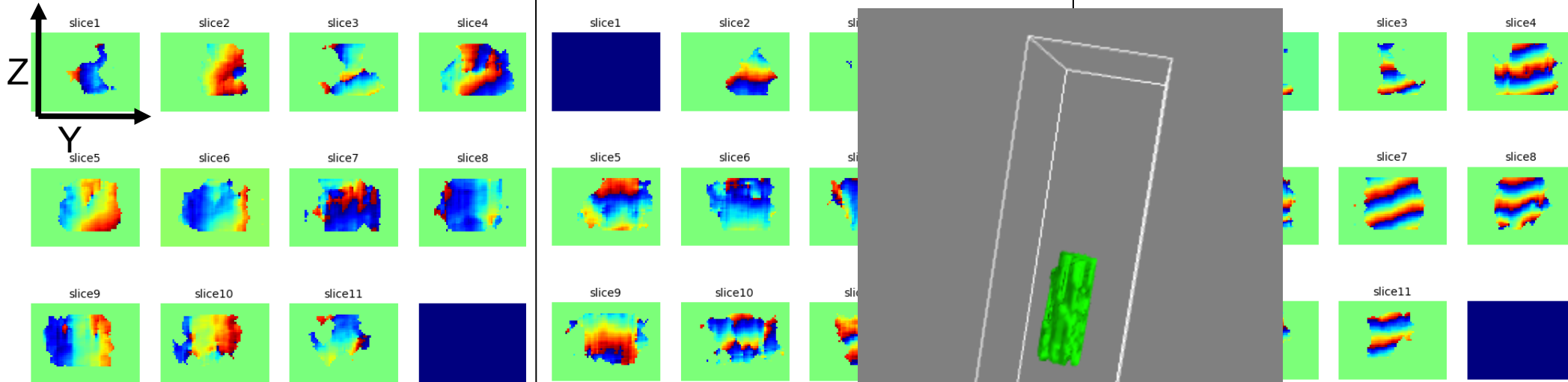


(111) Bragg peak
 $G \cdot u = G_{uz}(x,y,z)$

T=50deg

T=100deg

T=150deg



Progressive apparition

ion

Retrieval Procedure

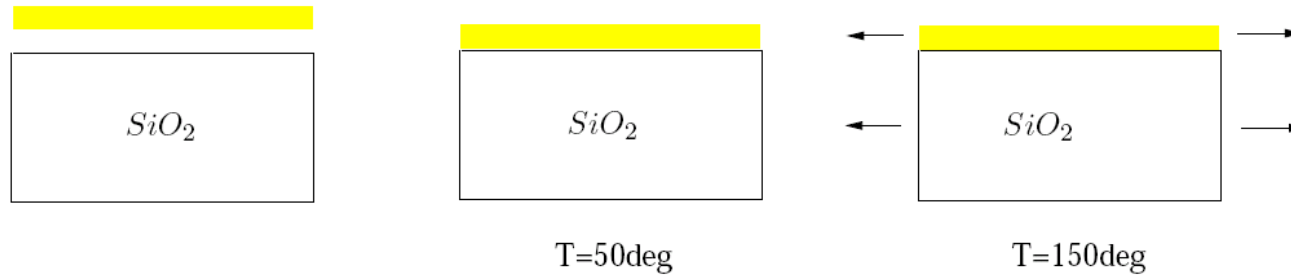
- Initial Support=cube estimated from fringe spacing
- 5 shrinkwraps
- Each Cycle $5 \cdot \{750ER + 250HIO\}$
- = 50 000 FFT (50 hours)



Could be local minima : complementary calculations under

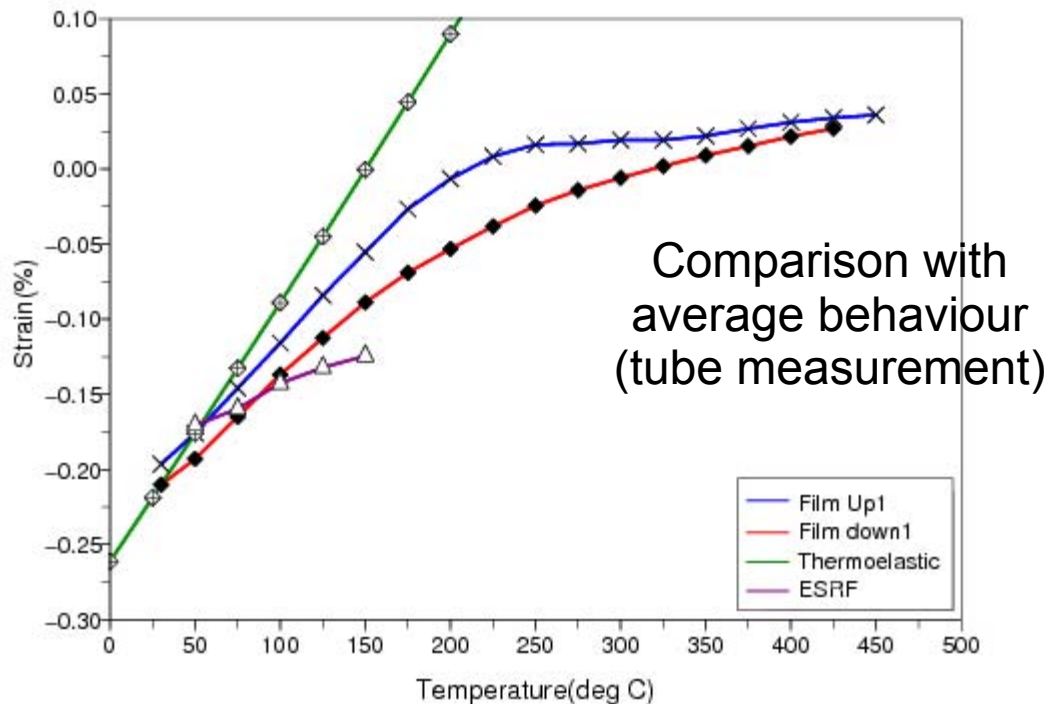
progress

Some average result



Thermo-elastic behaviour

$$\epsilon_{//} = - [\alpha_{Au} \Delta(T) - \alpha_{SiO_2} \Delta(T)]$$



Sample imperfections

- No perfect adhesion with substrat
- Joint grooving
- Incertitude on furnace calibration

Numerical aspect



Our cluster

- 62 processors
- GigE communication

Parallel computing

- 50 000 iterations of retrieval algorithm for a $384 \times 384 \times 31$ pixel tensor == 20 days of calculation on a classical computer
- Parallel is needed for large tensor of data
- Helpful for testing numerous parameters

Open Source Solutions

- FFTW 2.1.5 library C/C++/Fortran functions (MPI)
- New 3.3alpha version
- Python distarray Package under developement (B. Granger)

A parallel FFT need significant communication between processors

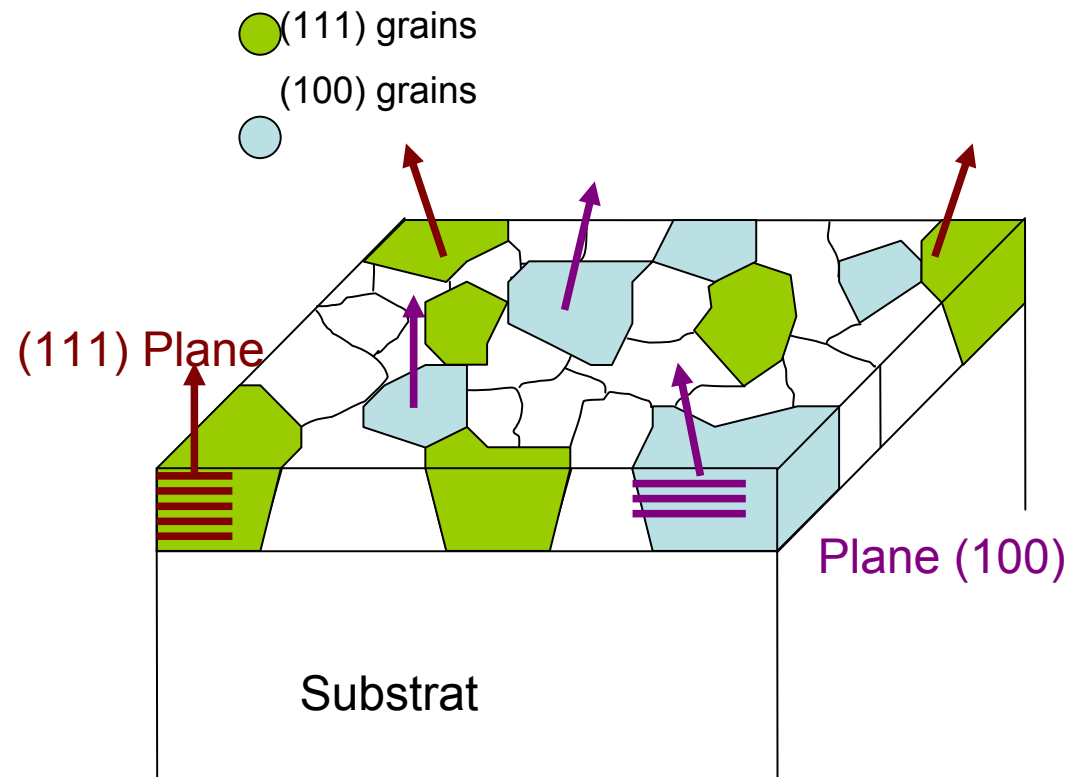
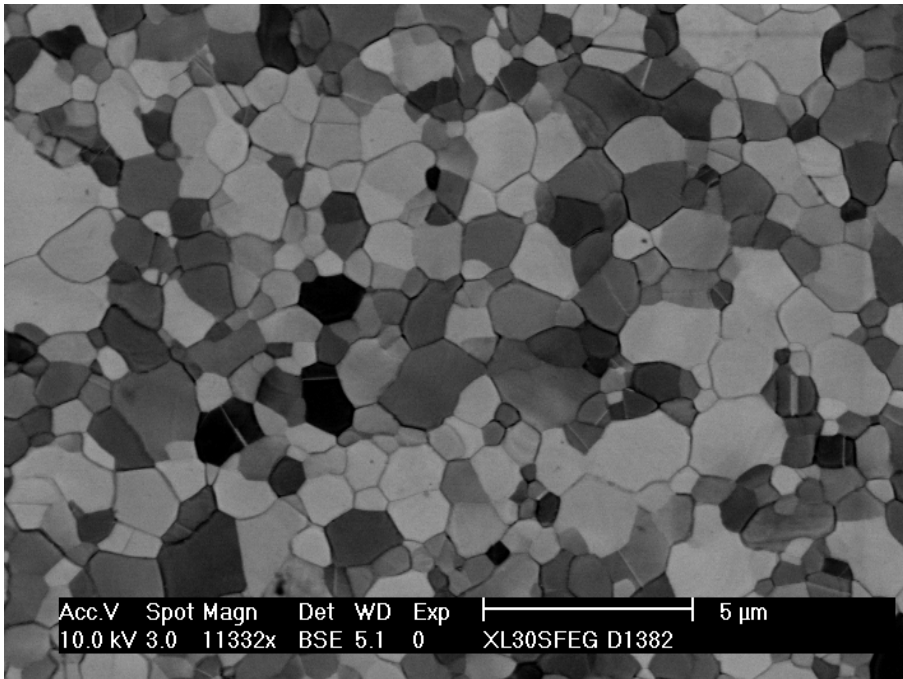


Benefits from multicore

Polycrystalline Thin Film

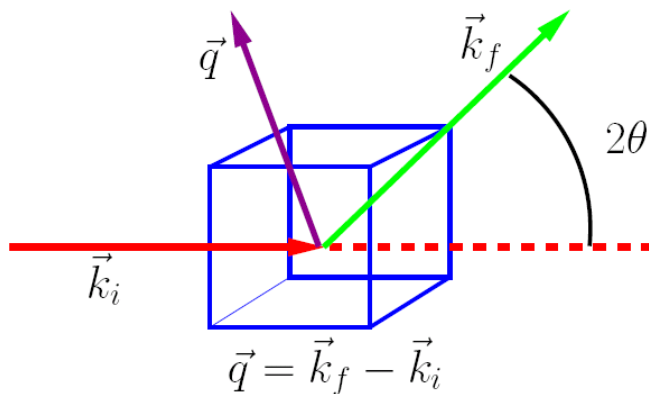
Main Features

- › Grain size distribution
- › Texture
- › Orientation inhomogeneities
 - › In plane
 - › Out-of plane



Kinematic theory of diffraction

Intensity distribution in reciprocal space



Complex Magnitude

$$A(\mathbf{q}) \propto \sum_n f_n(\mathbf{q}) \exp(i\mathbf{q}\mathbf{r}_n)$$

$$\mathbf{r}_n = \mathbf{R} + \mathbf{u} \quad \text{Displacement field}$$

$$A(\mathbf{q}) = \sum \left(f_n \exp(i\mathbf{q}\cdot\mathbf{u}(\mathbf{r})) \right) \exp(i\mathbf{q}\cdot\mathbf{R})$$

Diffracted Intensity

$$I(\mathbf{q}) = \left| \sum_n f_n(\mathbf{q}) \exp(i\mathbf{q}\mathbf{r}_n) \right|^2$$



$$I(\mathbf{q}) \propto |TF \{ \rho(\mathbf{r}) \cdot \exp(i\mathbf{G}\cdot\mathbf{u}(\mathbf{r})) \}|^2$$

Takagi approximation

S. Takagi. *J. Phys. Soc. Jpn*, 1969.

$$A(\mathbf{q}) \propto TF \{ \rho(\mathbf{r}) \cdot \exp(i\mathbf{G}\cdot\mathbf{u}(\mathbf{r})) \}$$